

Van der Waals forces in graphitic nanostructures

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*QMC in the Apuan Alps IV
Vallico Sotto, 28th July 2008*



History

Johannes Diderik [van der Waals](#) (1837-1923):

→ Postulation of intermolecular attraction in gases and liquids

Fritz [London](#):

→ Unified treatment of “dispersion forces” in noble gases

[Eisenschitz und London, Z. Physik 60, 491 (1930)]

Henk [Casimir](#) and Dirk Polder

→ Proposed force between metal plates in vacuum

“The Influence of Retardation on the London-van der Waals Forces”

[Phys. Rev. 73, 360 (1948)]



Definitions of terms

Inter-molecular forces: anything **except** covalent or ionic bonds

- A) dipole-dipole force (two permanent dipoles)
[more general: multipole forces]
- B) hydrogen bond
- C) induced dipole force (permanent dipole/induced dipole)
- D) dispersion forces (instantaneous dipole-dipole force)

London force

→ synonym for “dispersion force” (sometimes including induced dipole force)

van der Waals force

→ synonym for “intermolecular force”
(sometimes synonym for dispersion force)

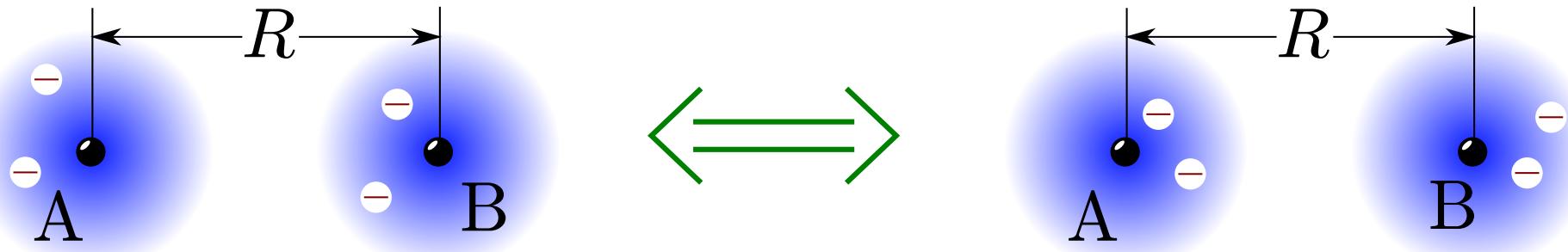
Casimir force

→ dispersion force between solids (continuum theory)



Dispersion force between atoms

⇒ long-distance correlations between electron positions within unpolar atom



$$E_{AB}^{\text{disp}} \approx -\frac{3\alpha^A \alpha^B I_A I_B}{4(I_A + I_B)} R^{-6}$$

α^A, α^B polarizability

I_A, I_B ionization potential

R interatomic distance

obtained via QM-multipole expansion [Eisenschitz und London, Z. Phys. 60, 491 (1930)]



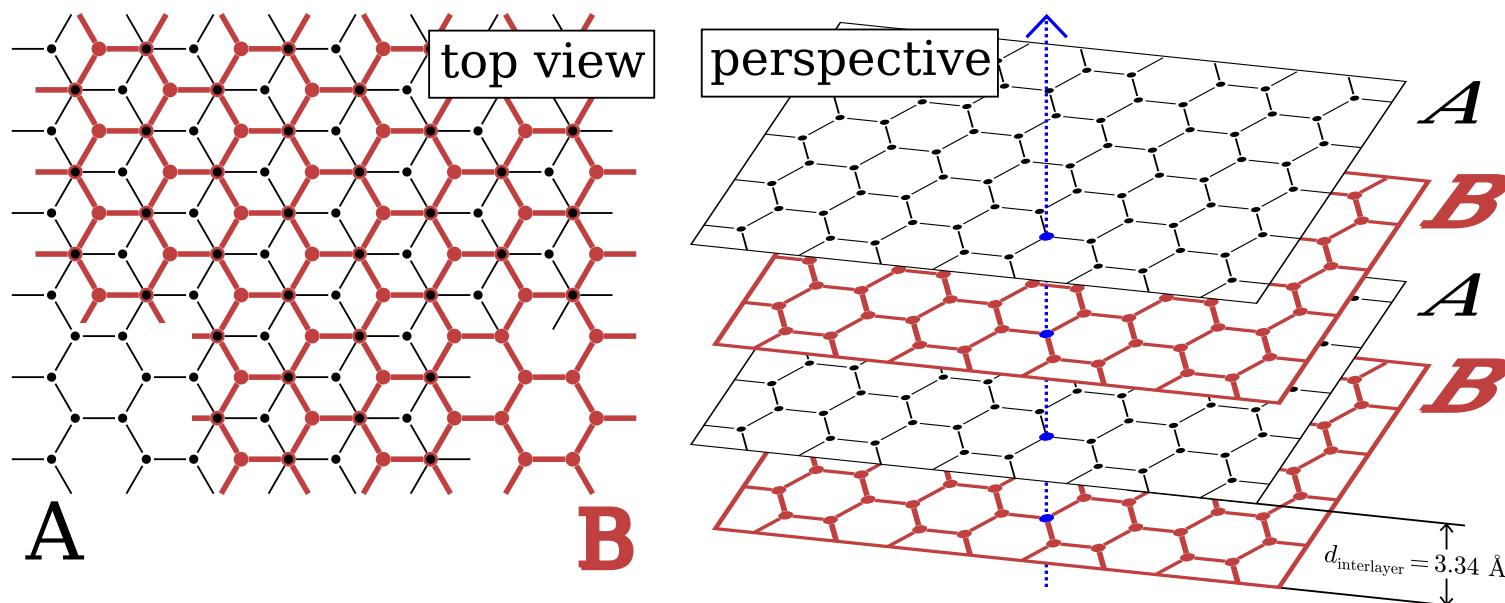
Casimir force

Macroscopic bulk description of force between polarizable media.

- depends on *dynamic polarizability* and *geometry* of media
- not additive (!)
- in case of metal bodies in vacuum
 ⇒ elegant computation via *vacuum energy* of intermediate space
- always attractive for symmetric combination of media
(e.g. metal-vacuum-metal, air-liquid-air, etc.)
- reduced with increasing temperature
- reduced by relativistic retardation important for longer distances



Graphite



intralayer bond length:

atomization energy

intralayer isotrop. elastic constant: $C_{11} + C_{22} = 1240 \text{ GPa}$

$$d_{CC} = 1.4196 \text{ \AA}$$

$$E_{\text{at}} = 7.374 \text{ eV/atom}$$

interlayer distance:

exfoliation energy:

interlayer elastic constant:

$$d_{\text{interlayer}} = 3.335 \text{ \AA}$$

$$E_{\text{ex}} = 35 \dots 52 \text{ meV/atom}$$

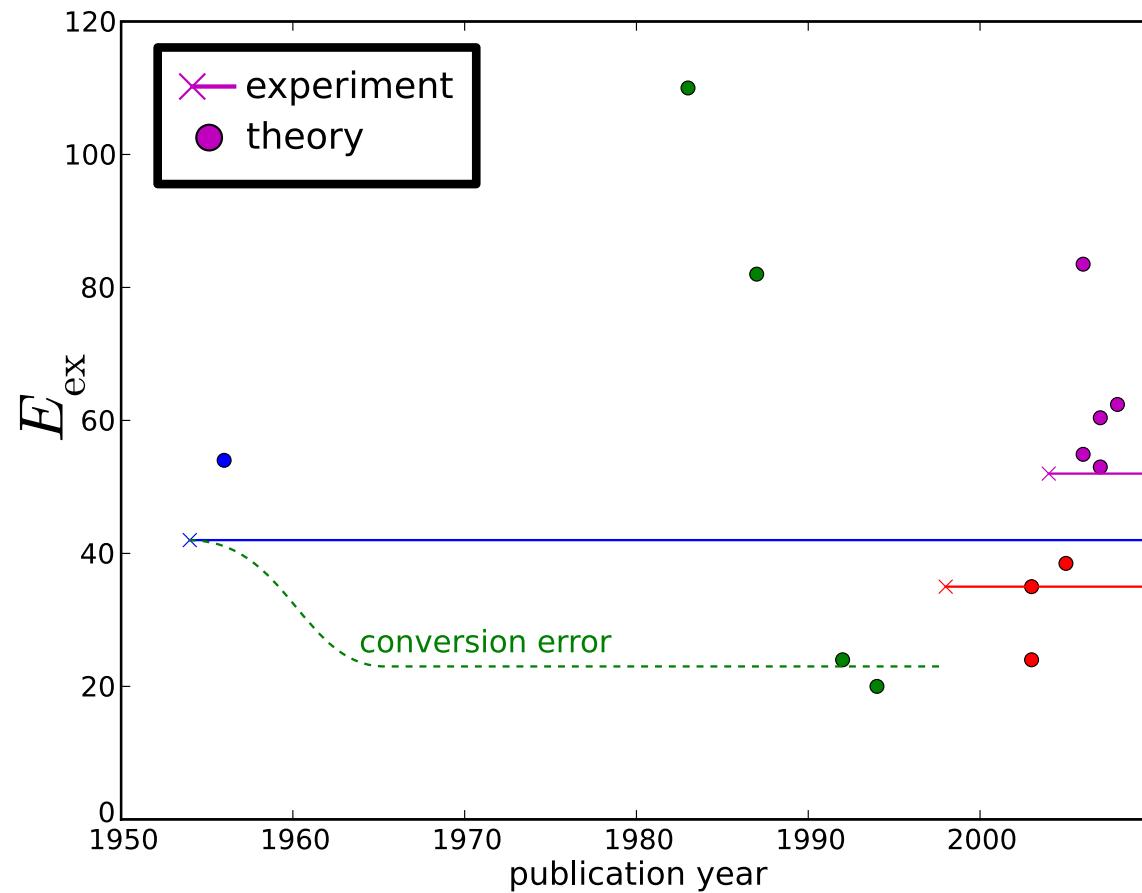
$$C_{33} = 36.5 \text{ GPa}$$

Theoretical attempts on interlayer energetics

	Method	$d_{\text{interlayer}} \text{ (\AA)}$	$E_{\text{ex}} \left(\frac{\text{meV}}{\text{atom}} \right)$	$C_{33} \text{ (GPa)}$
Brennan (1952)	LCAO + vdW	exp. input	11 / 172	39 / 11
Girifalco (1956)	lattice summation		54	
DiVincenzo (1983)	DFT + Thomas Fermi	5.6	110	
Yin (1984)	DFT-LDA	7.05 ± 0.7	~ 13.6	54 ± 20
Jansen (1987)	all elec. augmented PW	6.83 ± 0.06		56 ± 9
Schabel (1992)	DFT-LDA	6.72	24	24.3
Charlier (1994)	DFT-LDA	6.60	20	
Telling (2003)	DFT-LDA	6.70	35	
Rydberg (2003)	layered DFT-vdW	7.52	24	13
Zhechkov (2005)	DFTB + a posteriori vdW	6.76	38.5	
Mounet (2005)	GGA (at exp. lattice const.)	exp. input		45
Donchev (2006)	QMPFF	6.972	54.9	40.6
Ortmann (2006)	GGA + semiemp. vdW	6.69	83.5	41.7
Ziambaras (2007)	general DFT-vdW	7.18	53	27
Hasegawa (2007)	DFT + semiemp. corr.	exp. input	60.4	exp. input
Gould (2008)	LDA/GGA + semiemp. corr.	exp. input	62.4 / 59.7	exp. input



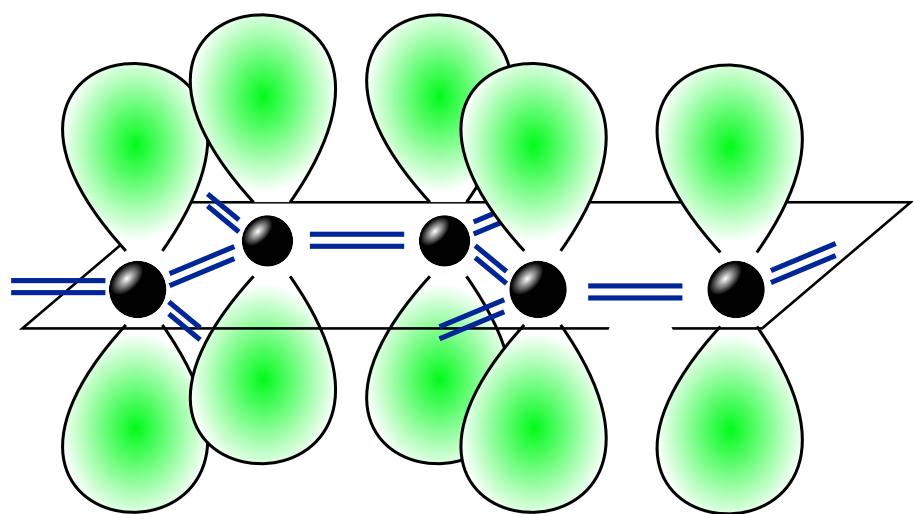
Correlation between experiment and theory



exfoliation energy:

- 42 meV/atom [Girifalco 1952: wetting by organic liquids]
- ~ 20 meV/atom → various incorrect conversions
- 35 meV/atom [Benedict 1998: collapse of nanotubes]
- 52 meV/atom [Zacharia 2004: polycyclic aromatic hydrocarbons]

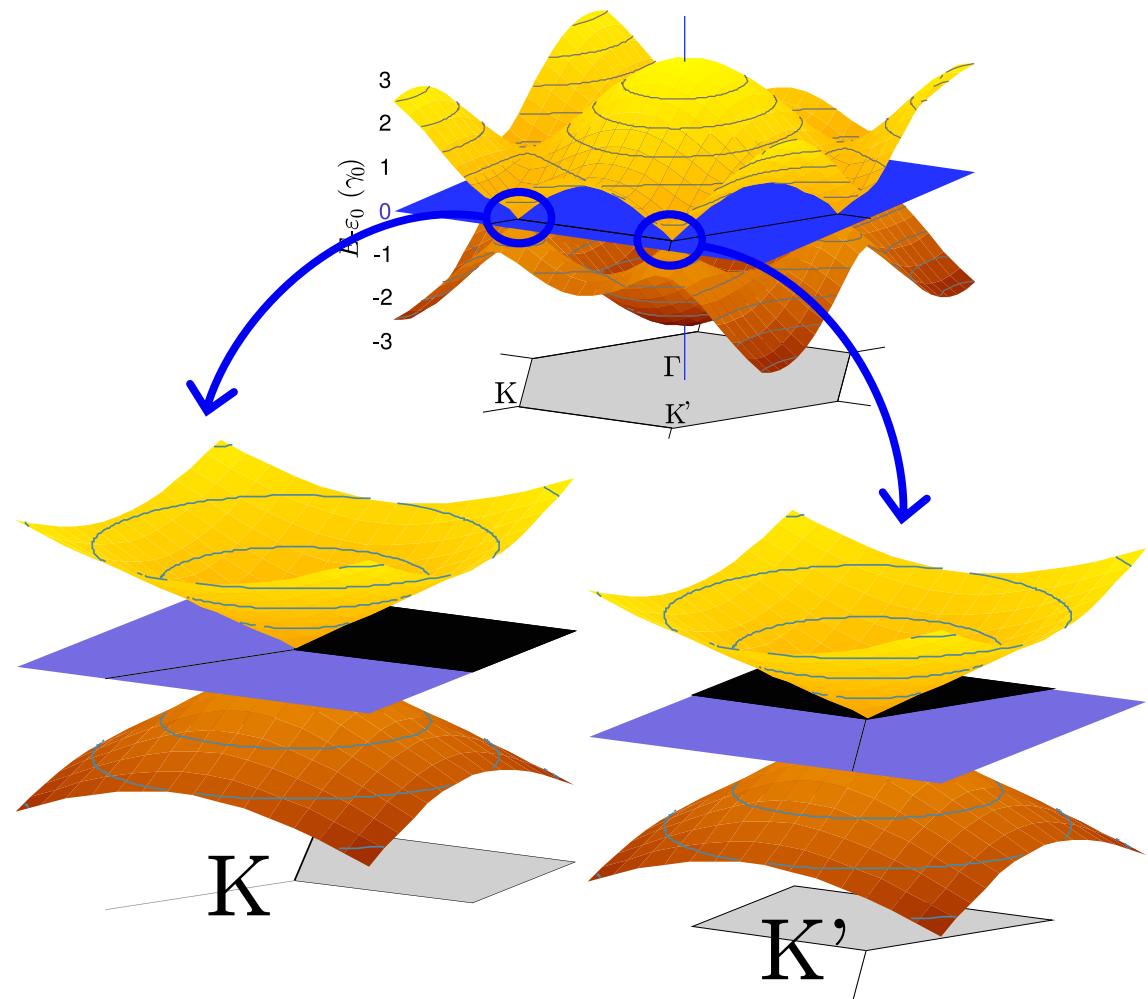
π -conjugated electronic structure of graphene



in-plane sp^2 -orbitals
→ strong σ -bonds

remaining p_z -orbitals
→ semi-metallic π -bands

π -band structure of graphene:



Dispersion forces for thin layers

TABLE I. Asymptotic vdW energy of parallel structures. K and D_0 are constants.

System	Present	Standard
1D metals ^a	$-D^{-2}(\ln(KD))^{-3/2}$	$-D^{-5}$
1D insulators [9]	$-D^{-5}$	$-D^{-5}$
2D metals [10,11]	$-D^{-5/2}$	$-D^{-4}$
π -conjugated layers ^a	$-D^{-3}$	$-D^{-4}$
1 metallic, 1 π layer ^a	$-D^{-3} \ln(D/D_0)$	$-D^{-4}$
2D insulators [6]	$-D^{-4}$	$-D^{-4}$
<i>Thick</i> metals or ins. [11]	$-D^{-2}$	$-D^{-2}$

^{a*} Denotes new derivations given here.

[from Dobson, White and Rubio, Phys. Rev. Lett. **96**, 073201 (2006)]



Graphite and QMC

[Fahy et al., Phys. Rev. B **42**, 3503 (1990)]

- VMC with nonlocal pseudopotential
 - precision insufficient for vdW interactions

[Prendergast et al., Phys. Rev. B **66**, 155104 (2002)]

- VMC with inhomogeneous e-e-Jastrow term
 - no binding energy extracted
 - large number of parameters (3000 for 3x3x3 super cell)

Our approaches (work in progress!!)

vdW-tailored Jastrow term

DMC with finite-size corrections

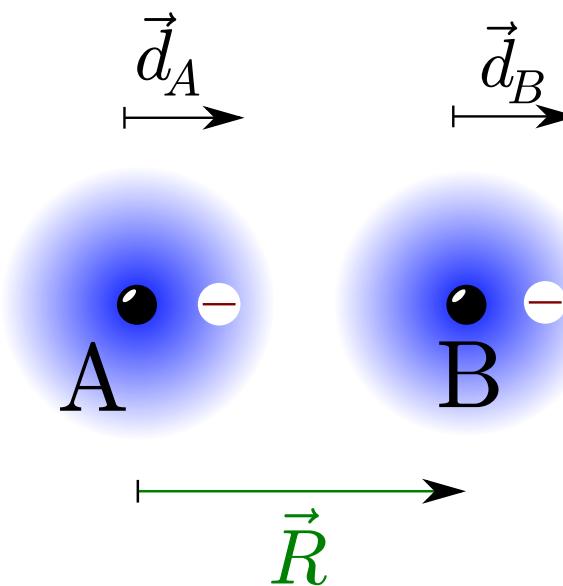


The Jastrow-D-term

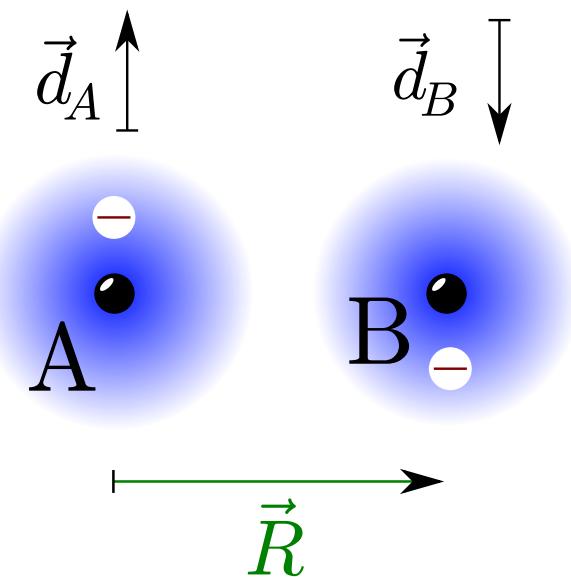
Inspired by classical dipole-dipole interaction:

$$E = \frac{1}{4\pi\epsilon_0 R^3} (\mathbf{d}_A \cdot \mathbf{d}_B - 3(\mathbf{d}_A \cdot \mathbf{R})(\mathbf{d}_B \cdot \mathbf{R}))$$

longitudinal dipoles



attractive transverse dipoles



The Jastrow-D-term

Inspired by classical dipole-dipole interaction:

$$E = \frac{1}{4\pi\epsilon_0 R^3} (\mathbf{d}_A \cdot \mathbf{d}_B - 3(\mathbf{d}_A \cdot \mathbf{R})(\mathbf{d}_B \cdot \mathbf{R}))$$

Two independent D-terms:

$$D_{\parallel} = \sum_{I < J} \sum_{i,j} (\mathbf{r}_{Ii} \cdot \mathbf{R}_{IJ})(\mathbf{r}_{Jj} \cdot \mathbf{R}_{IJ}) f_{\parallel}^{IJ}(r_{Ii}, r_{Jj})$$

$$D_{\perp} = \sum_{I < J} \sum_{i,j} [(\mathbf{r}_{Ii} \cdot \mathbf{r}_{Jj}) R_{IJ}^2 - (\mathbf{r}_{Ii} \cdot \mathbf{R}_{IJ})(\mathbf{r}_{Jj} \cdot \mathbf{R}_{IJ})] f_{\perp}^{IJ}(r_{Ii}, r_{Jj})$$

f_{\parallel}^{IJ} and f_{\perp}^{IJ} : scalar functions with cutoff, analogous to U term
→ optimized independently for each pair of atoms (reduced by symmetry)

⇒ implemented, but yet to be tested ...

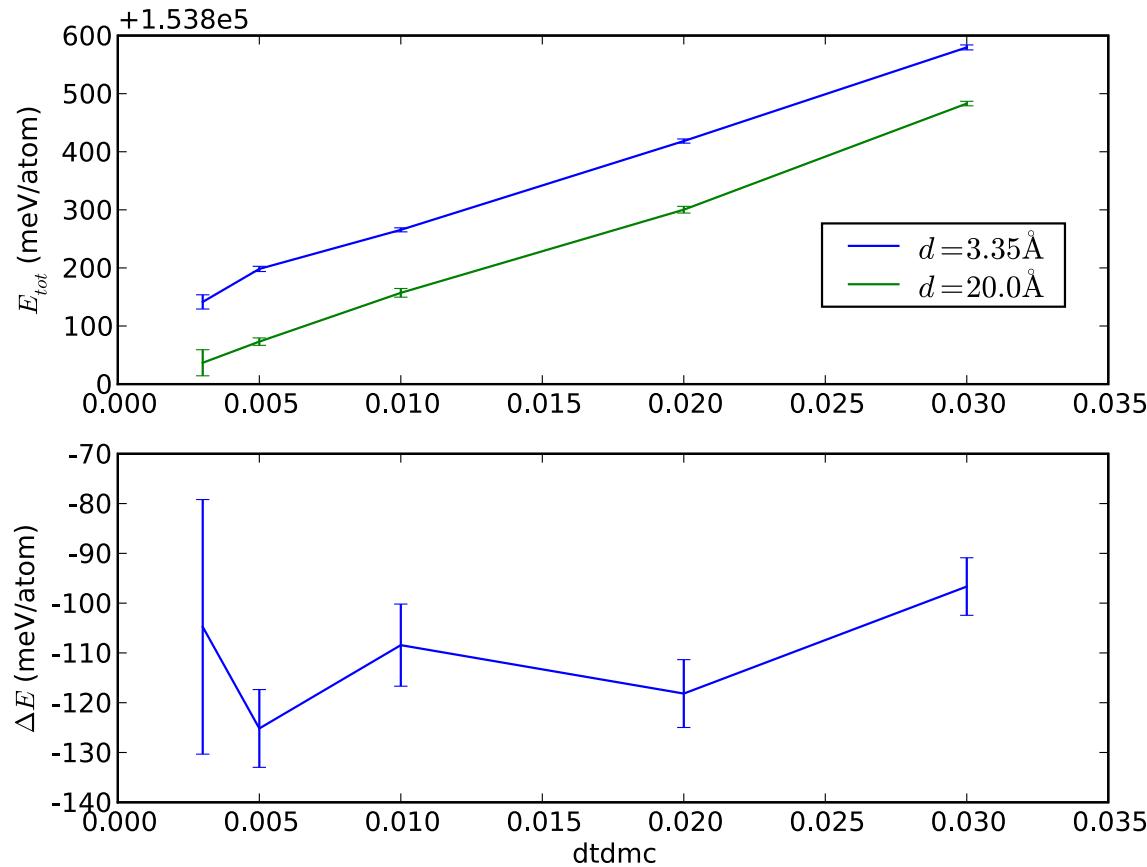


Graphite interlayer binding with CASINO

- CASTEP wave function (plane-wave → blip)
- using pseudo-potentials ⇒ 4 electrons per C atom
- primitive cell containing two graphene layers with two atoms each
⇒ 16 electrons per primitive cell
- starting out with 3x3x1 simulation cell
⇒ 144 electrons in simulation cell
- Jastrow terms: U , χ , F
→ $N_u = N_\chi = 6$, $N_{f-\text{ee}} = N_{f-\text{eN}} = 2$
→ fixed cutoffs $L_u = 4$ a.u., $L_\chi = 3$ a.u., $L_f = 2$ a.u.
→ varmin-linjas optimization



DMC time-step and CPU cost analysis



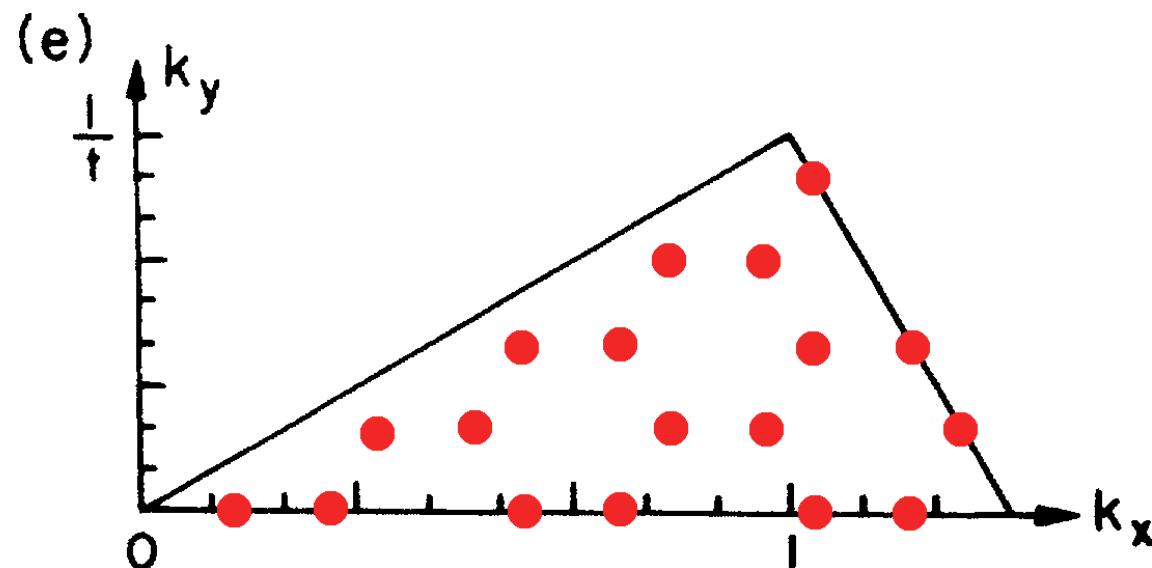
→ 64 nodes á 100 configs
→ 1000 time steps data collection
→ $5 \times 64 = 320$ CPU hours

→ correlation time: ~ 2 a.u.
→ using $dtdmc=0.01$



Twist averaging

18 twist angles according to special points in hexagonal 2D Brillouin zone



[Cunningham, Phys. Rev. B 10, 4988 (1974)]



Preliminary results

super cell	E_{ex} (meV/atom)	CPU hours
$3 \times 3 \times 1$	102 ± 6	2×1000
$3 \times 3 \times 2$	68 ± 5	2×5000
$4 \times 4 \times 1$	53 ± 4	2×5000
<i>experiment</i>	$35 \dots 52$	

Next steps

- more careful Jastrow optimization
- finite size correction for kinetic energy
- Ewald \leftrightarrow MPC
- check error from finite k-grid in DFT
- localize blip wfn (not yet implemented for complex wfn)
- larger systems and more smaller and intermediate system sizes



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